

# Synthesis and Sensing Properties to $\text{NH}_3$ of Hexagonal $\text{WO}_3$ Metastable Nanopowders

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$\text{WO}_3$  is an important kind of wide-bandgap semiconducting metal oxides, which has a very promising property in gas-detection behavior. It has several polymorphs with triclinic, monoclinic, orthorhombic structures being the stable forms of this oxide. However, by a method known as *acid precipitation*, new metastable open crystalline forms with hexagonal structure have been successfully synthesized. The nanopowders were characterized by SEM, TEM, and XRD, and their sensing response to reducing gases ( $\text{NH}_3$ ) was measured and compared to monoclinic  $\text{WO}_3$ , showing a much better sensing property of the hexagonal  $\text{WO}_3$ .

**Keywords** Acid precipitation; Ammonia; Hexagonal; Metastable; Monoclinic; Nanopowder; Open structure; SEM; Sensitivity; Sensor; Synthesis; TEM; Thin film; Tungsten oxide; XRD.

## 1. INTRODUCTION

Metal oxides are polymorphic compounds and controlled chemical processing may stabilize oxide polymorphs that would otherwise be energetically unstable. Recent studies by the authors' group, [1–3] led to the hypothesis that the ability for selective detection of a particular gaseous analyte in the presence of interfering gas mixtures (i.e., sensor selectivity) is largely determined by the chosen crystalline polymorph (specific crystallographic phase) of a stoichiometric and pure metal oxide used for sensing. Transition metal oxide such as  $\text{MoO}_3$  and  $\text{WO}_3$  were used as model systems in those studies.

Orthorhombic ( $\alpha$ -phase)  $\text{MoO}_3$  was found to exhibit high specificity to the detection of ammonia and amines.  $\text{MoO}_3$  has a layered structure with (010) basal plane that is built up of double chains of edge-sharing  $[\text{MoO}_6]$  octahedral units connected through vertices. Thus the  $\alpha$ -phase is selective to ammonia and highly sensitive to amines (which are moderate bases) because their sensing mechanism involves the reduction of  $\text{MoO}_3$  and the formation of ordered phases (reaction-based sensing process) as it has confirmed in the works of Gouma using XPS studies, [2, 3]. It is suggested that transition metal oxides with “loosely-bound” and open structures are necessary in order to achieve selective amine detection, as they enable the reaction of lattice oxygen with the gas and provide easy mechanisms for accommodating the off-stoichiometric M:O ratio [4]. Issues of structural stability due to the high volatility of this material, however, limit its use in high temperature sensing applications. The search for alternative materials that are isostructural with  $\text{MoO}_3$  identified  $\text{WO}_3$  as promising candidates.

$\text{WO}_3$  has been used to detect various gases due to its excellent semiconducting properties. Since Shaver reported the first  $\text{WO}_3$  gas sensor for the detection of hydrogen [5], there has been numerous reports on  $\text{WO}_3$  sensors for detecting  $\text{NO}_2$  and  $\text{NH}_3$  and other gases such as ozone,  $\text{CO}$ ,  $\text{CH}_4$ , and  $\text{H}_2\text{S}$ , esp. during the last 5 years. Xu et al. [6] have reported their studies of selective detection of  $\text{NH}_3$  over  $\text{NO}$  in combustion exhausts by using Au and  $\text{MoO}_3$  doping on  $\text{WO}_3$  elements. The effects of dopants [7], substrate dependence [8], calcining and operating temperatures [9, 10], stoichiometry and microstructure [11], film thickness [12] on  $\text{WO}_3$  based films were studied respectively by different groups. The enhanced response of Pt, Au Cr [13], Cu or V [14]-doped  $\text{WO}_3$  nanocrystalline powders to  $\text{NH}_3$  were also studied. Tomchenko et al. [15] designed and fabricated semiconducting metal oxide sensor array for the selective detection of combustion gases. Teoh et al. [16] and Kim et al. [17] have even developed room-temperature  $\text{WO}_3$  based gas sensors. Recently, multi-wall carbon nanotube modified  $\text{WO}_3$  thin films were obtained and proved to highly improve gas sensing properties [18]. In addition, tungsten oxide sensing layers on highly ordered nanoporous alumina template has been synthesized by Khatko et al. [19].

$\text{WO}_3$  has several polymorphs with triclinic, monoclinic, orthorhombic, and tetragonal structures being the stable forms of this oxide, and a metastable hexagonal structure ( $h\text{-WO}_3$ ). However, all these previous works used stable structures to fabricate the sensors and they haven't discussed the structure influence on the gas sensing properties, esp. the sensing properties of  $h\text{-WO}_3$  haven't been reported.  $h\text{-WO}_3$  was first synthesized and reported in 1979 by Figlarz's group [20] by the dehydration of a tungsten oxide hydrate compound ( $\text{WO}_3 \cdot 1/3\text{H}_2\text{O}$ ). Its structure is built up of  $\text{WO}_6$  octahedra arranged in layers normal to the hexagonal c-axis, forming hexagonal tunnel structures [21]. Furthermore,  $h\text{-WO}_3$  has attracted attention as an electrochromic material [22, 23] thus that it might be possible to produce sensors

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that change color in the presence of reducing gases, such as CO.

In this paper, by a method called acid precipitation, new metastable open h-WO<sub>3</sub> crystalline forms can be synthesized. This work refers to the ammonia sensing properties of two polymorphs of WO<sub>3</sub>, the stable monoclinic and the metastable hexagonal phase.

## 2. EXPERIMENTAL

To synthesize the hexagonal structured WO<sub>3</sub> powders, 1.17 g of Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O of analytical grade was dissolved in 17 ml of water and the solution was cooled to 10°C. To this 8.4 ml of normal hydrochloric acid solution (analytical grade, 18% in excess of equimolar reaction) cooled to the same temperature was added in one dose. The mixture was put back into the refrigerator and allowed to stay for about 20 h. After this time the whole mixture turned to a whitish gel. Then 110 ml of water was added to the vessel and the gel and water were lightly stirred manually. After centrifuging the supernatant liquid was removed. Then 130 ml of water was added to the precipitate and the steps of light manual stirring, centrifuging and removal of supernatant liquid were repeated several times to obtain H<sub>2</sub>WO<sub>4</sub>·H<sub>2</sub>O, the precursor of final h-WO<sub>3</sub> powders. H<sub>2</sub>WO<sub>4</sub>·H<sub>2</sub>O suspensions were passed to hydrothermal dehydration, carried out in Parr acid digestion bombs at autogeneous pressure at 125°C ± 5°C.

Dehydration under air: furnace temperature: 300–330°C, annealing time: 90 min. [24].

To prepare for the h-WO<sub>3</sub> gas sensor, 0.1 g of such h-WO<sub>3</sub> powders were weighed into 3 ml of heptanol and the suspension was ultrasonically stirred for at least 1 h. Two drops of the suspension were removed to drip on an Au electrode-coated Al<sub>2</sub>O<sub>3</sub> substrate. The substrate is left at the room temperature for 1 h until the solution evenly spread on the substrate. Then the substrate was dried at 75°C for 10 min to remove the liquid. Such process was repeated for 3 times and then a uniform h-WO<sub>3</sub> thin film gas sensor was successfully prepared.

## 3. RESULTS AND DISCUSSION

The color of h-WO<sub>3</sub> powders is gray, different from that of common monoclinic WO<sub>3</sub> (γ-WO<sub>3</sub>), which is yellowish green. The scanning electron microscopic (SEM) morphology of h-WO<sub>3</sub> powders is shown in Fig. 1(a).

In this image there are two typical shapes of WO<sub>3</sub> powders in the product. One shape is nanoparticles and the other is nanorods. The two shapes of powders are mixed together. Fig. 1(b), (c) are transmission electron microscopic (TEM) images of WO<sub>3</sub> nanoparticles and nanorods, respectively. The nanoparticles are about 10–50 nm in diameter while the nanorods are about 30–100 nm in diameter and 100–300 nm in length.

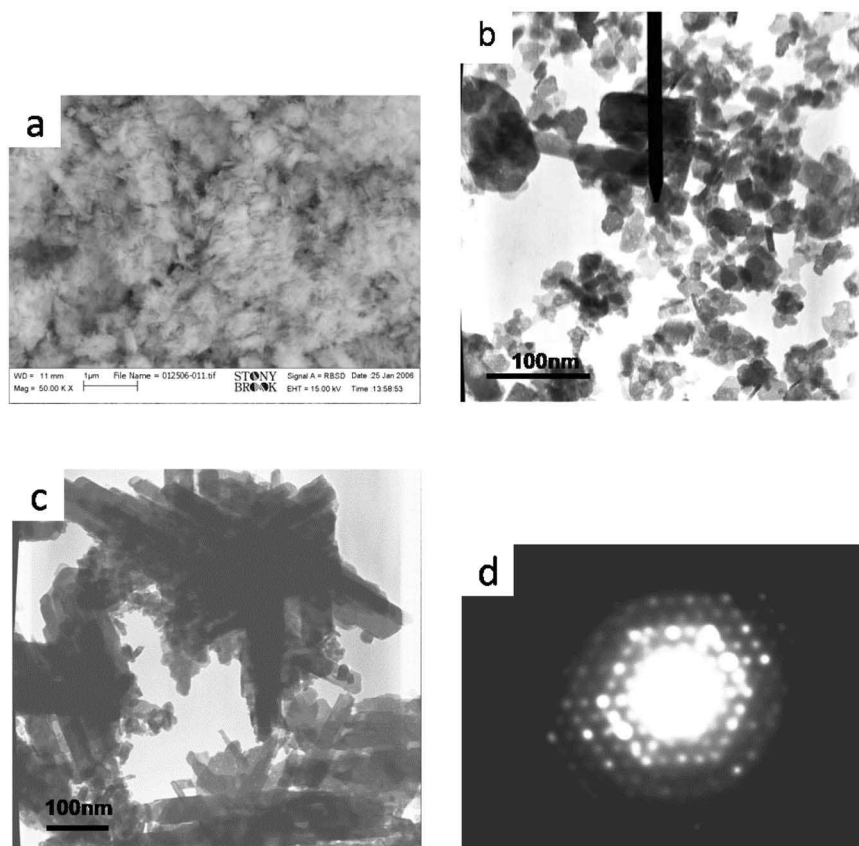
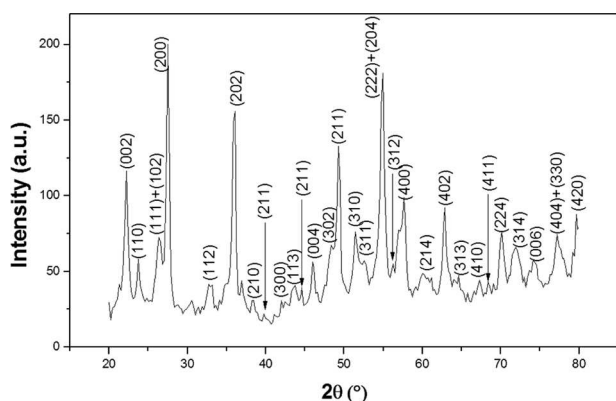


FIGURE 1.—Morphology and structure of h-WO<sub>3</sub> powders. a) SEM image showing two shapes; b) TEM image of WO<sub>3</sub> nanoparticles; c) TEM image of WO<sub>3</sub> nanorods; d) SAED pattern of WO<sub>3</sub> powders showing a hexagonal structure.

FIGURE 2.—XRD result of h-WO<sub>3</sub> powders.

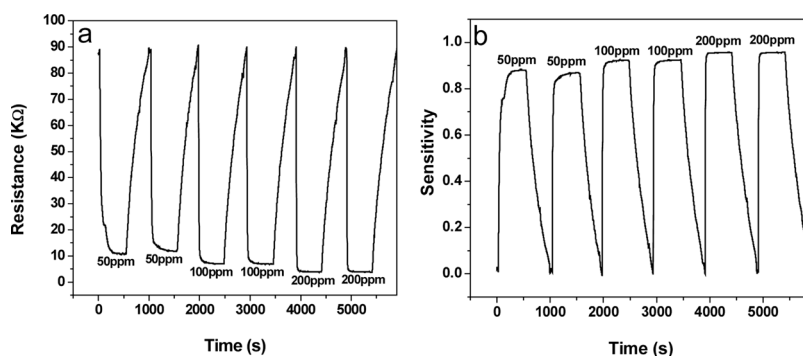
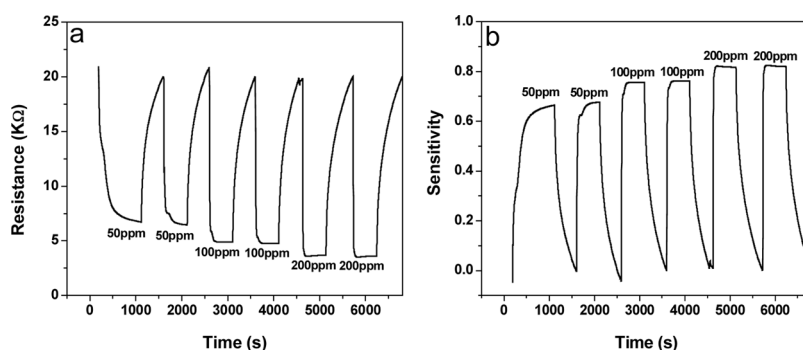
Although the WO<sub>3</sub> nanoparticles and nanorods are different in shape, they share a same structure. Fig. 1(d) is selected area electron diffraction (SAED) pattern of the WO<sub>3</sub> powders. It shows that the powders have a hexagonal structure. Such result is confirmed by the following X-ray diffraction (XRD) result.

The XRD result of the metastable powders is shown in Fig. 2. According to the result, the product is in a high purity and can be indexed as a primitive hexagonal structure with the lattice parameter of  $a = 0.7324$  nm and  $c = 0.7662$  nm, compared to the standard data file (JCPDS file No.: 852459).

To measure the sensing property of the h-WO<sub>3</sub> nanopowders, we prepared a thin film sensor using these powders as source material on Au electrode-coated Al<sub>2</sub>O<sub>3</sub> substrate (shown in the “experimental method” chapter). The sensing test was carried on at 400°C and we used ammonia (NH<sub>3</sub>) as the testing gas. When the concentration of NH<sub>3</sub> is 50 ppm, 100 ppm, and 200 ppm, the resistance has lowered down from 90 kΩ to about 11 kΩ, 7 kΩ and 4 kΩ (shown in Fig. 3(a)) and the sensitivity is up to 88%, 92%, and 96% (shown in Fig. 3(b)), respectively. This result indicates an outstanding sensing response of h-WO<sub>3</sub> nanopowders to NH<sub>3</sub> reducing gas. In addition, the sensitivity at a certain concentration of NH<sub>3</sub> is reproducible, showing that the sensing property of the h-WO<sub>3</sub> nanopowders is stable which can be regarded as an excellent candidate for detecting NH<sub>3</sub> gas.

To compare the sensing property of h-WO<sub>3</sub> to other structured WO<sub>3</sub>, we also prepared γ-WO<sub>3</sub> thin film gas sensor using the same method. The powders have similar morphology and sizes with h-WO<sub>3</sub>, but have different structures. The sensing response and sensitivity of γ-WO<sub>3</sub> are shown in Fig. 4. At a NH<sub>3</sub> concentration of 50 ppm, 100 ppm, and 200 ppm, the sensitivity of γ-WO<sub>3</sub> is 67.5%, 75.5%, and 82%. Although it indicates a good sensing response of γ-WO<sub>3</sub> to NH<sub>3</sub>, it is much lower than that of h-WO<sub>3</sub> nanopowders.

Therefore, we can conclude that h-WO<sub>3</sub> powders have a much better sensing property than γ-WO<sub>3</sub> nanopowders. As mentioned in the Introduction chapter, h-WO<sub>3</sub> has a

FIGURE 3.—The (a) response and (b) sensitivity to NH<sub>3</sub> gas of h-WO<sub>3</sub> powders at 400°CFIGURE 4.—The (a) response and (b) sensitivity to NH<sub>3</sub> gas of γ-WO<sub>3</sub> powders at 400°C

loosely-bound and more open layered structure. When it is exposed to  $\text{NH}_3$ , it is believed that  $\text{NH}_3$  tend to remove oxygen from lattice and form ordered phases. Such reaction leads to the higher sensing properties of h- $\text{WO}_3$  to  $\text{NH}_3$ .

#### 4. CONCLUSION

$\text{NH}_3$  selective gas sensor based on tungsten oxide nanopowder films were successfully synthesized using acid precipitation method. The product has a new hexagonal structured metastable phase rather than common stable phases of tungsten oxides, such as monoclinic etc. The synthesized h- $\text{WO}_3$  sensor has a very high sensitivity to  $\text{NH}_3$  gas, up to 88%, 92%, and 96% responding to 50 ppm, 100 ppm, and 200 ppm of  $\text{NH}_3$ , respectively, which is much higher than that of  $\gamma$ - $\text{WO}_3$  sensor which has similar morphology and sizes. It is believed that it is the more open structure of h- $\text{WO}_3$  that improves its sensitivity compared to other stable phases.

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