

1 Upconversion Particle as a Local Luminescent Brownian Probe: A 2 Photonic Force Microscopy Study

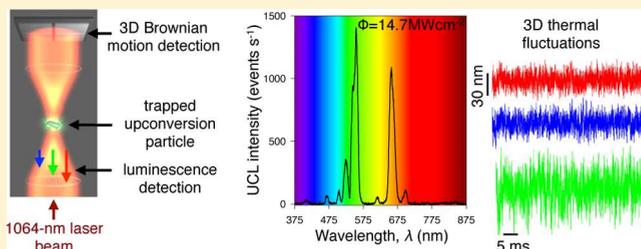
3 Flavio M. Mor,* Andrzej Sienkiewicz, László Forró, and Sylvia Jeney

4 Laboratory of Nanostructured and Complex Matter Physics (LPMC), Institute of Condensed Matter Physics (ICMP), Faculty of
5 Basic Sciences (FSB), Ecole Polytechnique Fédérale de Lausanne (EPFL), CH-1015 Lausanne, Switzerland

6  Supporting Information

7 **ABSTRACT:** Near-infrared (NIR) light sensitive lanthanide-
8 doped NaYF₄ upconversion particles (UCPs) are gaining
9 increasing attention as local probes in biomedical applications.
10 Here, we implemented a photonic force microscope (PFM) to
11 manipulate and study the optical properties of trapped single
12 UCPs, β -NaYF₄:Yb,Er. In particular, we focused on the
13 mechanisms of the optical trapping of nonspherical UCPs of
14 different sizes, in the range 0.5–2 μ m, as well as on their
15 upconversion photoluminescence (UCL) properties under
16 excitation with the strongly focused laser beam ($\lambda = 1064$ nm)
17 of the PFM, operating at power densities up to 14.7 MW cm⁻². A careful analysis of UCL under such conditions points to three
18 emission peaks at 469, 503.6, and 616.1 nm, which were enhanced by the high laser power density. The analysis of Brownian
19 motion was used to quantify the thermal fluctuations of the particle inside the optical trap as well as the particle sizes and optical
20 forces acting in the two dimensions perpendicular to the optical axis. A steep dependence of UCL as a function of the particle
21 diameter was found for UCPs having sizes smaller than the focal spot (~ 900 nm) of the NIR laser.

22 **KEYWORDS:** β -NaYF₄:Yb,Er particle, optical trapping, optical tweezers, nonlinear optics, upconversion photoluminescence,
23 Brownian motion



24 **L**ife sciences have a constantly growing need for novel
25 methodological approaches suitable to investigate the
26 intracellular environment with increased time and spatial
27 resolutions. Optical near-field probes, such as laser-illuminated
28 metal tips, uncoated/metal-coated tapered optical fibers, and
29 nanoemitters, such as single molecules or nanoparticles, have
30 attracted increasing attention as key components of high-
31 resolution microscopes.

32 In parallel, super-resolution fluorescence microscopy, operat-
33 ing in the far-field regime and overcoming the light diffraction
34 limit, has markedly improved imaging resolution. Although,
35 near- and far-field optical approaches drastically improve image
36 resolution, they still present numerous drawbacks, which
37 include technical difficulties related to probe preparation
38 (near-field) or potentially damaging light intensities (far-field).

39 The aforementioned shortcomings of high-resolution optical
40 microscopy can be overcome to a great extent by photonic
41 force microscopy (PFM). PFM employs a strongly focused
42 near-infrared (NIR) laser light to hold a dielectric or metallic
43 particle as a local probe. Such an optical trap enables confining
44 a mesoscopic particle and track its three-dimensional (3D)
45 thermal fluctuations in the surrounding environment, e.g., a
46 viscous liquid. Therefore, besides offering 3D near-field
47 imaging, PFM also reports on other important quantities
48 concerning the local mechanical properties, including force and
49 viscosity as well as dynamical properties of the surrounding
50 medium. This additional information is derived from the careful

51 analysis of the so-called Brownian motion of the trapped single-
52 particle probe that collides with thermally activated surround-
53 ing molecules.

54 At present, the most promising nonlinear, light-emitting
55 nanoprobes that have been shown to be optically manipulable
56 are nanowires made of perovskite alkaline niobates (e.g.,
57 KNbO₃)^{1,2} or indium phosphide.³ These nanowires are able to
58 convert the continuous infrared (IR) light energy of the optical
59 trap into one sharp visible line at half the excitation wavelength
60 by a second-harmonic generation (SHG) process. Long KNbO₃
61 nanowires have potential application for near-field probing.¹ In
62 particular, it was shown to be possible to excite organic
63 fluorescent molecules, such as nucleic acid binding POPO-3
64 dyes, by a direct energy transfer from the trapped KNbO₃
65 nanowire emitting the SHG light.¹ However, SHG requires
66 relatively high levels of IR power⁴ and challenges remain
67 concerning preparation of SHG probes having controlled sizes
68 and shapes suitable for optical manipulation.^{1,2}

69 Light-upconversion systems, which are capable of converting
70 low-energy IR or NIR incident photons to higher energy
71 emission, are increasingly gaining attention as novel probes for
72 analytics in biological environments, quantitative microscopy,
73 and bioimaging.⁵ In particular, lanthanide-doped upconversion
74 luminescent nanoparticles (UCNPs) display the unique

Received: May 28, 2014

75 property of emitting visible light following photoexcitation with
76 NIR laser light.⁶ Such an upconversion photoluminescence
77 (UCL) mechanism is generally not available for existing
78 endogenous and exogenous fluorophores, offering UCNPs
79 various distinctive characteristics for optical biomedical
80 imaging, such as minimal autofluorescence background, large
81 anti-Stokes shifts (up to 500 nm), narrow emission bandwidths,
82 high resistance to photobleaching, nonblinking, less light
83 scattering, and deeper tissue penetration.^{7,8} Additional
84 advantages over other light-emitting probes are their higher
85 quantum yield and a tunable emission wavelength.

86 UCNPs and their microsized upconversion counterparts
87 (UCPs) are now becoming commercially available in a wide
88 range of shapes and sizes.^{9,10} Typically, characterization of their
89 photophysical properties is done in bulk,^{9–12} and only a few
90 studies of the properties of single upconversion particles have
91 been reported.^{13–16}

92 Recently, the feasibility of 3D optical trapping of small-sized
93 UCNPs (~26 nm in diameter) was demonstrated by Haro-
94 González et al.¹⁷ In the present work, we developed a
95 methodology based on PFM and high-frequency optical
96 trapping interferometry^{18,19} to simultaneously study photo-
97 physical and morphological properties of single UCPs. We
98 implemented two types of commercially available highly
99 luminescent UCPs based on β -NaYF₄, which were nominally
100 doped with Yb and Er. These two types of UCPs were
101 synthesized by two different technological pathways, i.e., the
102 flame-fusion method and hydrothermal standard Schlenk
103 technique, which resulted in two markedly different particle
104 morphologies, that is, random and well-defined (hexagonal
105 plate) shapes, respectively.

106 The rationale behind investigating UCPs having different
107 morphologies, yet revealing similar photo-optical properties
108 under experimental conditions of PFM, was imposed by the
109 adopted procedure of calibration of the Brownian motion for
110 particles of arbitrary shapes. In particular, first, we developed
111 the calibration procedure for a particle having well-defined
112 morphology (hexagonal plate), and then, the protocol was
113 extended to UCPs of random shapes.

114 Our PFM experimental setup enabled us to investigate the
115 UCL properties of UCPs at the single-particle level, at very high
116 NIR laser power densities, up to 14.7 MW cm⁻², and revealed
117 strong peaks of visible light emission related to the
118 upconversion transitions within the electronic manifold of
119 Er³⁺. We also observed additional, well-distinguished, although
120 definitely much weaker UCL peaks, which were enhanced by
121 the high laser power density. Moreover, unlike in bulk studies
122 of UCPs, in PFM, interactions with neighboring particles could
123 be avoided, thus minimizing distortions in UCL spectra of
124 individual UCPs. Finally, the exact analysis of Brownian
125 fluctuations of trapped single UCPs made it possible to
126 evaluate the best-suited morphology of such particles for
127 applications as near-field luminescent Brownian probes.

128 ■ RESULTS AND DISCUSSION

129 To investigate the Brownian fluctuations of an optically trapped
130 particle with 1 μ s temporal and 1 nm spatial resolutions, we
131 used a custom-built PFM setup.²⁰ It consists of an ultrastable
132 inverted light microscope with a 3D sample positioning stage, a
133 1064 nm laser (IRCL-500-1064-S, CrystaLaser, USA) for
134 optical trapping, and a quadrant photodiode (QPD), model
135 G6849 from Hamamatsu Photonics, Japan, for high-resolution
136 as well as time-resolved position detection.

The PFM setup was modified to detect and measure
137 nonlinear optical effects emitted by a single optically trapped
138 particle excited directly by the 1064 nm trapping beam in
139 continuous wave (CW) mode. In order to perform high-
140 sensitivity photoluminescence detection, a spectrofluorometer
141 (USB2000-FL-450, Ocean Optics Inc., USA) and an electron
142 multiplying charge-coupled device (EMCCD, iXon^{EM}+897,
143 Andor Technology, United Kingdom) camera were added
144 (Figure 1).

145 f1

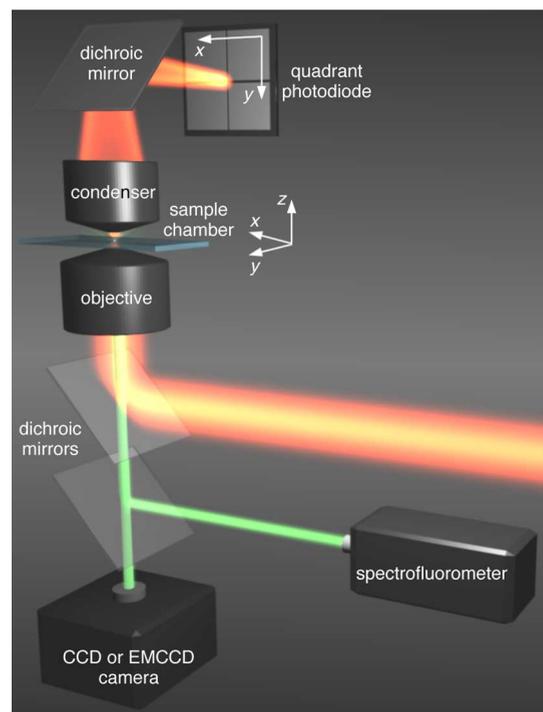


Figure 1. Schematic layout of the laser light pathways for particle trapping (orange light path) and for UCL detection (green light path). The Brownian motion of the trapped UCP was monitored with a QPD. The UCL spectra were acquired with a fiber-coupled spectrofluorometer. The trapped particle can be visualized with a CCD or an EMCCD camera. The NIR laser power can be varied by selected neutral density filters (see Supporting Information for more details).

Two types of lanthanide-doped samples with similar Yb/Er
146 ratios were studied: (i) polycrystalline NaYF₄:2.2%Yb,2%Er
147 UCPs having random shapes, edges, and sizes ranging from
148 hundreds of nanometers to a few micrometers⁹ (Figure 2a) and
149 (ii) the highly monodisperse, monocrystalline NaYF₄:20%
150 Yb,2%Er plates of hexagonal shape, with edge lengths of ~230
151 nm and a thickness of ~120 nm¹⁰ (Figure 2b). UCPs of the
152 first type, PTR550/F, were bought from Phosphor Technol-
153 ogy Ltd., Stevenage, England. These UCPs comprise
154 polycrystalline NaYF₄:2.2%Yb,2%Er UCPs ground from bulk-
155 grown crystals obtained by the flame-fusion method. UCPs of
156 the second type were purchased from Intelligent Solutions Inc.,
157 Princeton, NJ, USA. These UCPs comprise monocrystalline
158 NaYF₄:20%Yb,2%Er plates of hexagonal shape synthesized by
159 standard Schlenk techniques.¹⁰ The as-received random shape
160 UCPs were tip-sonicated to achieve stable suspensions in water,
161 whereas the monodispersed hexagonal shape UCPs were
162 dispersed in water using a sonication bath (see Supporting
163 Information for more details).
164

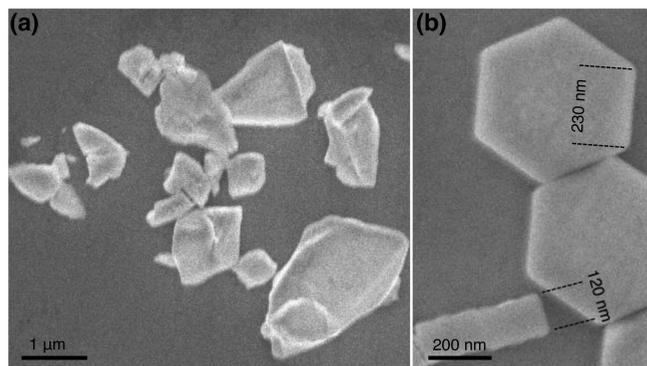


Figure 2. High-resolution scanning electron microscopy images of (a) random and (b) hexagonal shape UCPs. The detector for secondary electrons was placed 3 mm from the surface sample, and the energy of the excitation electrons was 3 keV for high-resolution imaging of the particles.

To study the UCL properties at the single-particle level, a dispersed aqueous solution of UCPs was first introduced into the sample chamber of our PFM. Next, the UCPs were allowed to sediment to the bottom surface. A single UCP was then selected by optical microscopy and picked up with the optical trap away from the bottom surface of the sample chamber. To make sure that we trapped a single UCP, during the experiment, we continuously checked the UCL spectra and monitored the Brownian motion. In particular, the amplitude of the UCL spectrum becomes higher when a second UCP particle enters the optical trap (see Supporting Information for more details). Finally, the UCL intensity spectrum of each optically trapped UCP excited by the NIR trapping laser was recorded. Figure 3a shows typical UCL intensity spectra for a random and hexagonal shape UCP trapped at different laser power densities Φ_{laser} . The laser power density was calculated knowing the $1/e^2$ radius $w_0 = 451 \text{ nm}^{21}$ and the laser power, P_{laser} , as $P_{\text{laser}}/(\pi w_0^2)$. The measurements were performed at room temperature. According to the model developed by E. J. G. Peterman et al. for polystyrene and silica microspheres,²² one can expect the temperature increase in the fluid adjacent to the particle to be $\sim 2 \text{ }^\circ\text{C}$ for the maximum laser power (94 mW) we used (see Supporting Information for more details). The UCL emission spectrum of a single optically trapped UCP exhibits a multicolored spectrum with nine distinguishable emission bands centered around 408.6, 469, 503.6, 525.4, 549.2, 616.1, 657, 697.8, and 841.9 nm. The wavelength for each emission band does not change because it is dopant- but not shape-dependent. Interestingly, the green light emitted by the trapped hexagonal plate is less quenched than for random shape particles (Figure 3a, black line).

The UCL intensity spectrum (Figure 3a) originates from the upconversion process between Yb^{3+} as a donor and Er^{3+} as an acceptor.²³ Peaks located at 408.6, 525.4, 549.2, and 657 nm are respectively attributed to the radiative transitions from the $^2\text{H}_{9/2}$, $^2\text{H}_{11/2}$, $^4\text{S}_{3/2}$, and $^4\text{F}_{9/2}$ excited states to the $^4\text{I}_{15/2}$ ground state of Er^{3+} (Figure 3b). These transitions were also visible in bulk measurements performed at low laser power densities.⁹ In contrast, the blue emission peaks around 469 and 503.6 nm matching the intrastate transitions $^2\text{K}_{15/2} \rightarrow ^4\text{I}_{13/2}$ and $^4\text{G}_{11/2} \rightarrow ^4\text{I}_{13/2}$, and the red peaks at 616.1 and 697.8 nm corresponding to the intrastate transitions $^4\text{G}_{11/2} \rightarrow ^4\text{I}_{11/2}$ and $^2\text{H}_{9/2} \rightarrow ^4\text{I}_{11/2}$ respectively, have not been observed in bulk measurements performed using much lower laser power densities.^{9,10}

Previously, the emission peaks at 697.8 and 841.9 nm have also been reported for similar particles measured at moderate power densities up to $\sim 300 \text{ W cm}^{-2}$.²⁴ In addition, investigations at the single-particle level have revealed the peak at 697.8 nm for a laser power density of $\sim 625 \text{ kW cm}^{-2}$.¹³ Thus, our measurements indicate that the three intrastate transitions at 469, 503.6, and 616.1 nm are exclusively visible due to a marked enhancement of their respective amplitudes by high laser power densities (up to 14.7 MW cm^{-2}) of our PFM setup. Interestingly, as can be seen in Figure 3b, the intensity of the NIR peak located at 841.9 nm, which matches three different intrastate transitions corresponding to $^4\text{G}_{9/2} \rightarrow ^4\text{F}_{9/2}$, $^2\text{H}_{9/2} \rightarrow ^4\text{I}_{9/2}$, and $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{13/2}$, is also enhanced.

Even though we associated the UCL peaks at 469 and 503.6 nm with intrastate transitions of the 4f electron manifold of the Er^{3+} ions, they might also originate from impurities. However, available data on the purity of the studied UCPs do not suggest the presence of trace amounts of other lanthanides, which might contribute to these UCL peaks.^{9,10,25}

UCPs were exposed to variable laser power densities to evaluate their UCL emission. To this end, one representative and relatively large UCP of random shape ($\sim 1.2 \text{ }\mu\text{m}$) was trapped by the laser beam of the PFM, and its photoluminescence was measured as a function of laser power density, Φ_{laser} . Figure 3c shows the emitted UCL power, P_{UCL} , of the five major peaks (469, 525.4, 549.2, 657, and 697.8 nm) as a function of Φ_{laser} (see Supporting Information for the calculation of P_{UCL}). We chose a rather large UCP to assess over a broad laser power range the otherwise too weak light blue and dark red emission bands, centered around 469 and 697.8 nm, respectively. The strong laser power densities reached in the tightly focused laser beam of the optical trap allowed studying the highest energy electronic state excitable through the upconversion process (Figure 3b). As can be seen in Figure 3a and c, the UCL intensity increases with Φ_{laser} . Similar behavior of increasing UCL at very high laser power densities was observed by Gargas et al.¹⁶ The lowest power density ($\Phi_{\text{laser}} = 0.0867 \text{ MW cm}^{-2}$) corresponds to the minimal laser power necessary for trapping, which essentially depends on the size of the particle. It is worth noting that the laser power densities employed in this experiment were much higher than those reported in similar studies so far.

In principle, UCL is excited by the sequential absorption of N photons according to the relation $P_{\text{UCL}} \propto \Phi_{\text{laser}}^N$. The values of N were derived from fits of the log–log plots of $P_{\text{UCL}}(\Phi_{\text{laser}})$ (Figure 3c, black lines).^{24,26} At very low Φ_{laser} , far below trapping conditions, $2 < N < 3$ is typically observed in bulk, indicating a mixed two- or three-photon process.^{24,26} At the high excitation densities necessary for trapping ($0.07^{-2} \lesssim \Phi_{\text{laser}} \lesssim 1^{-2}$), we observe that N is reduced to ~ 1 due to saturation of the UCL process, except for the light blue emission band, which still exhibits a three-photon upconversion process, with $N \approx 3$ (Figure 3c). Further increasing the trapping strength yields $N < 1$, thus pointing to UCL quenching due to a lack of phonons.^{26,27} Nevertheless, even under such extreme conditions after 2 min of illumination at the highest power density, $\Phi_{\text{laser}} = 14.7 \text{ MW cm}^{-2}$, the UCP remained stable and only very low bleaching with no blinking was observed (see Supporting Information). This is in contrast to quantum dots (QDs)²⁸ and organic dyes.²⁹

Further manipulation and characterization of a single UCP were achieved in the optical trap, which was linearly polarized in the xy -plane perpendicular to the beam axis z . As the laser

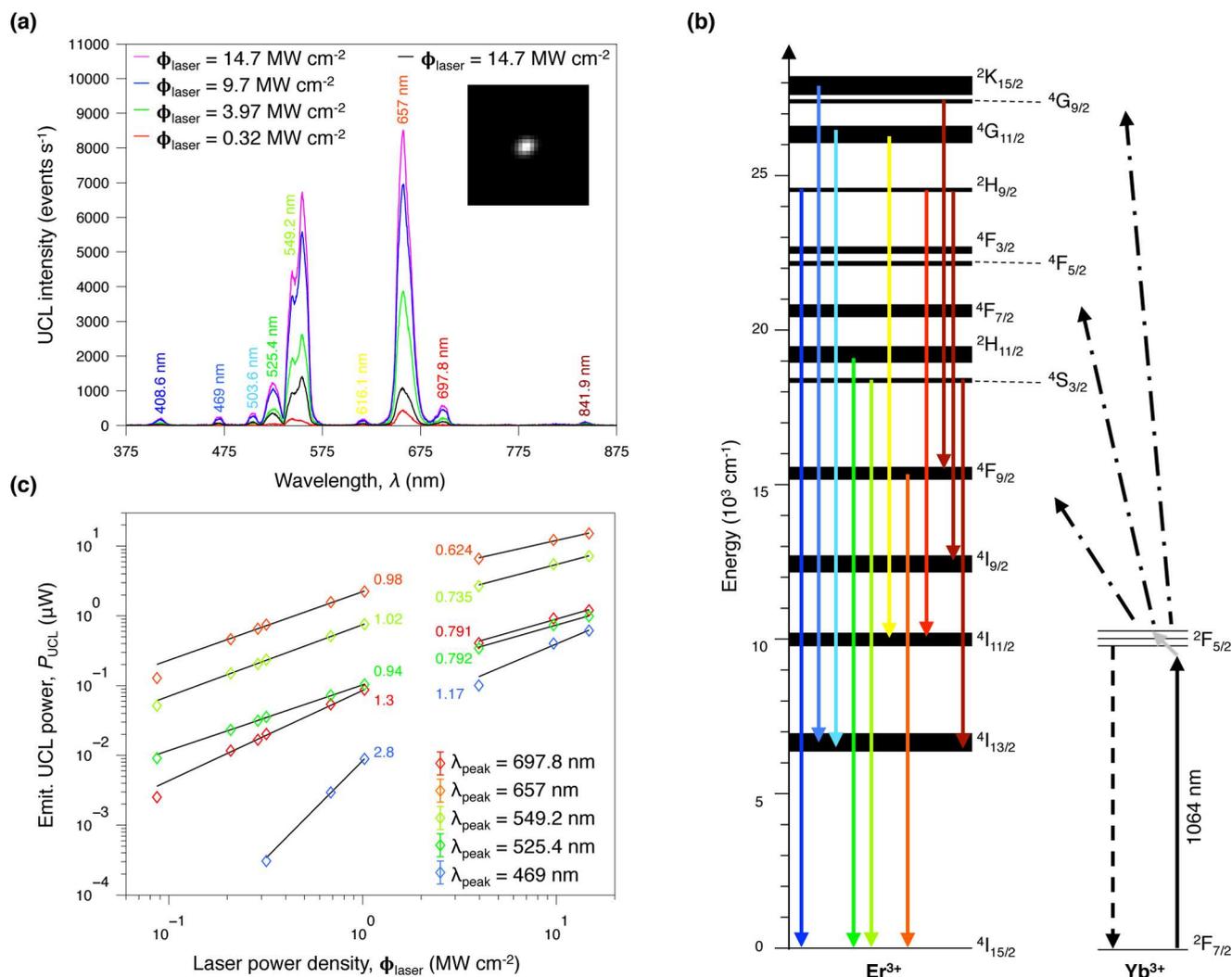


Figure 3. (a) Evolution of the UCL spectra from a single UCP as a function of the laser power density, Φ_{laser} , trapped at 0.32 (red trace), 3.97 (green trace), 9.7 (blue trace), and 14.7 (magenta trace) MW cm^{-2} , for a random shape UCP, and at 14.7 (black trace) MW cm^{-2} for a hexagonal shape UCP. All spectra were recorded using integration time $\tau_{\text{int}} = 0.5$ s, and the particles were trapped in water. Inset: Photoluminescent EMCCD image of a single trapped hexagonal UCP. (b) Representation of the *f*-shell electronic levels of Yb^{3+} and Er^{3+} atoms, showing the energy transfer UC processes (indicated by three dash-dotted black arrows) responsible for photoluminescence after NIR excitation (indicated by a continuous black arrow) of the ${}^2\text{F}_{5/2}$ level of Yb^{3+} . A phonon (indicated by a continuous gray arrow) is needed to excite the ${}^2\text{F}_{5/2}$ level of Yb^{3+} . The black dotted line represents the radiative relaxation of Yb^{3+} . The nine colored arrows correspond to the peaks in the UCL spectrum shown in (a). (c) Log–log representation of the UCL emitted power from the representative random shape particle for its five major peaks. The corresponding exponents of the power law (N) for each emitted wavelength are shown adjacent to the plots. All spectra were recorded with integration times ranging from 0.05 to 5 s.

272 light carries linear and angular momentum that is transferred to
 273 the particle, the optical trap exerts not only a restoring force but
 274 also a restoring torque on the geometrically anisotropic
 275 particle.^{30,31} All particles studied here align spontaneously
 276 with the laser polarization in the *xy*-plane of the trap and
 277 exhibit only small-angle oscillations around their equilibrium
 278 orientation. The remaining Brownian fluctuations of the
 279 trapped UCP are detected by the QPD.

280 When a UCP is trapped, the large-angle rotational Brownian
 281 motion is strongly suppressed by the optical restoring torque
 282 and short-time fluctuations are dominated by the translational
 283 Brownian motion.³² Then, the hydrodynamic vortex, which
 284 develops by momentum exchange between the fluctuating
 285 particle and the surrounding fluid, influences also Brownian
 286 motion. For a spherical particle of radius R_s , the hydrodynamic
 287 vortex propagates around the particle in all three dimensions

with one characteristic time constant, $\tau_f = \rho_s R_s / \eta_f$ where ρ_f is
 288 the density and η_f the viscosity of the fluid.¹⁹ In contrast, for a
 289 nonspherical UCP, the *x*- and *y*-components of translational
 290 Brownian motion are anisotropic and yield two characteristic
 291 time constants, $\tau_{f,i} = \rho_i R_{e,i}^2 / \eta_f$ with $i = x, y$. $R_{e,x}$ and $R_{e,y}$
 292 correspond to the effective radii along directions *x* and *y*,
 293 respectively, which are defined by the axis of the QPD (Figure
 294 1).
 295

As shown in Figure 4a, the time constants $\tau_{f,x}$ and $\tau_{f,y}$ appear
 296 in the short-time power laws $(t/\tau_{f,i})^{-3/2}$ of the velocity
 297 autocorrelation function (VAF) along the *x*- and *y*-directions,
 298 respectively. The values of $\tau_{f,x}$ and $\tau_{f,y}$ are derived from the
 299 Brownian motion of the trapped UCP. At longer times, each
 300 VAF_{*i*} exhibits a zero-crossing, which is then followed by a quasi-
 301 exponential relaxation with a characteristic time $\tau_{K_i} = 6\pi\eta R_{e,i}/$
 302 K_i , due to the restoring forces of the optical trap, with stiffness 303

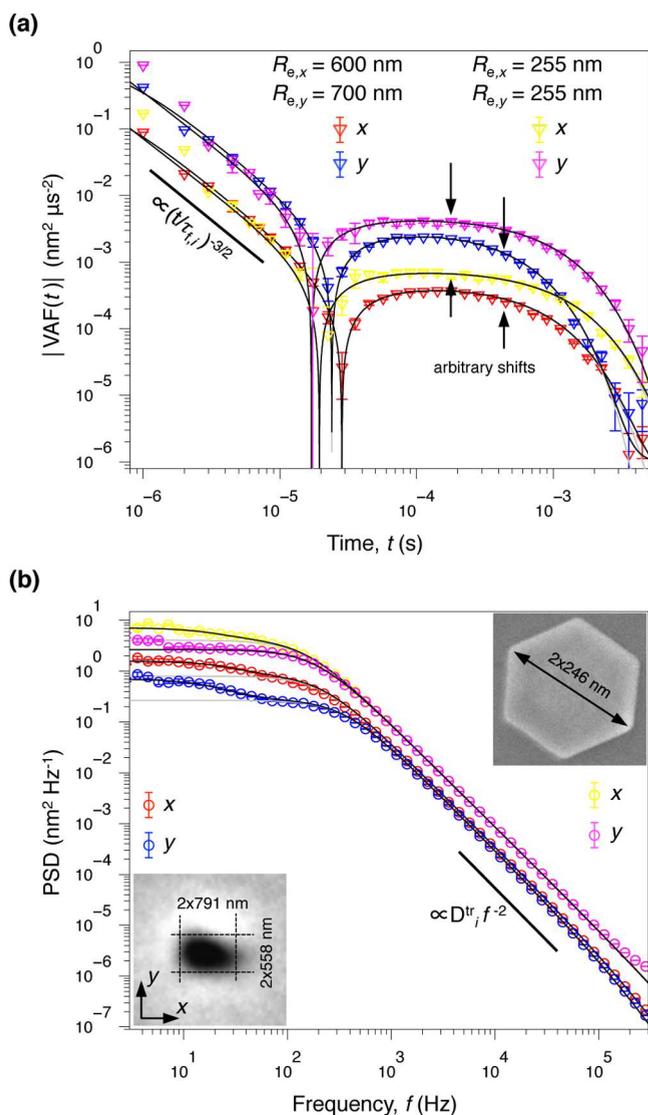


Figure 4. (a) Log–log plot of $|VAF|$ calibrated along the x - and y -directions for a random (red and blue data points) and hexagonal (yellow and magenta data points) shape UCP. (b) Log–log representation for the x - and y -components of the corresponding calibrated PSD. Insets: Images of the corresponding trapped UCPs. The gray lines indicate fitting based on the full hydrodynamic theory for a sphere,¹⁹ whereas the black lines take into account eqs 1 and 2 for the rotational motion. Both particles were trapped in water, at $\Phi_{\text{laser}} = 1.02 \text{ MW cm}^{-2}$ ($\rho_p = 4.21 \text{ kg m}^{-3}$, $\rho_f = 997.6 \text{ kg m}^{-3}$, and $\eta_f = 0.955 \text{ mPa s}$ at $T = 295 \text{ K}$). All data were blocked in 10 bins per decade, and error bars indicate the standard error of the mean values obtained from blocking.

304 K_i . Although the short-time fluctuations of the trapped UCP are
 305 primarily due to translational Brownian motion, the small-angle
 306 oscillations around its equilibrium orientation influence the
 307 long-time dynamics of the particle, as shown in the power
 308 spectral density (PSD) in Figure 4b for frequencies $f \lesssim 40 \text{ Hz}$.
 309 In principle, the influence of inertia of the fluid and the particle
 310 at these long times can be neglected for the rotational
 311 Brownian motion of an asymmetric particle.^{33,34} The rotational
 312 component of the VAF can be then expressed by a negative
 313 decaying exponential

$$VAF^{\text{rot}}(t) = -\frac{2D^{\text{rot}}}{\tau_{K_\phi}} e^{-t/\tau_{K_\phi}} \quad (1) \quad 314$$

and the PSD results in a Lorentzian function 315

$$PSD^{\text{rot}}(f) = \frac{D^{\text{rot}}}{\pi f^2 \left(1 + \left(\frac{f_{K_\phi}}{f} \right)^2 \right)} \quad (2) \quad 316$$

with the rotational diffusion constant, D^{rot} , and the time 317
 constant related to the optical torque, τ_{K_ϕ} . Expressions 1 and 2 318
 are solutions of the overdamped Langevin equation, which is a 319
 first-order equation with no dependence on the particle 320
 mass.^{33,34} 321

Under our experimental conditions, coupling between 322
 translational and rotational motions is low, and both 323
 contributions occur at well-separated time scales. As a 324
 consequence, our data can be described by a superposition of 325
 the translational contribution, given by Franosch et al.,¹⁹ and 326
 the rotational contributions expressed by eqs 1 and 2 (Figure 4, 327
 black lines). 328

At short times, only translational motion needs to be 329
 considered. Thus, adapting the method developed for a 330
 sphere¹⁹ to asymmetric particles is sufficient to fit data to 331
 theory along the x - and y -directions (Figure 4, gray lines). The 332
 method consists in simultaneously fitting the long-time tail in 333
 the short-time window of the VAF_i and the diffusive regime in 334
 the high-frequency window of the PSD_i , which is defined by the 335
 translational diffusion constant D_i^{tr} (Figure 4b). This yields 336
 effective radii $R_{e,x} = R_{e,y} = 255 \text{ nm}$ for a hexagonal plate, and $R_{e,x}$ 337
 $= 600 \text{ nm}$ and $R_{e,y} = 700 \text{ nm}$ for a typical random shape UCP. 338
 Both results match the apparent sizes measured either from the 339
 CCD camera calibrated image (Figure 4b, bottom left inset) for 340
 the random shape particle or by high-resolution scanning 341
 electron microscopy for the hexagonal plate (Figure 4b, top 342
 right inset). As expected, the apparent radius along the x -axis of 343
 the random shape particle, $R_{a,x} = 791 \text{ nm}$, is close to the 344
 effective radius obtained by fitting the y -component of the 345
 Brownian motion $R_{e,y} = 700 \text{ nm}$, and concomitantly $R_{a,y} \approx R_{e,x}$. 346
 Translational diffusion of the random shape particle along the 347
 x -axis is larger because $R_{a,y}$ is smaller (Figure 4b, $D_x^{\text{tr}} > D_y^{\text{tr}}$, red 348
 and blue data points). 349

Apparent sizes, $2R_{a,x}$ and $2R_{a,y}$, are defined as measurable 350
 sizes along the respective x - and y -axis from the CCD camera 351
 calibrated image for a given orientation of the UCP in the 352
 optical trap. In contrast, the orientation of a hexagonal shape 353
 plate remained unknown because it was too small to be clearly 354
 visualized with the CCD camera. Nevertheless, at high 355
 frequencies we found the translational diffusion constants D_x^{tr} 356
 $= D_y^{\text{tr}}$, whereas at low frequencies, $f \lesssim 40 \text{ Hz}$, data could not be 357
 fitted by the theory for a spherical particle (Figure 4b, magenta 358
 and yellow data points). 359

Through the detection of Brownian motion of a single UCP 360
 held in the optical trap, a connection between the size of the 361
 particle and the intensity of its photoluminescence can be 362
 made. In Figure 5, we observe an increase of the emitted UCL 363
 power, P_{UCL} , of major emission bands at 525.4, 549.2, 657, and 364
 697.8 nm (depicted in Figure 3a and b) with increasing UCP 365
 sizes. For an average effective radius $\langle R_e \rangle = (R_{e,x} + R_{e,y})/2$, 366
 below $\sim 400 \text{ nm}$, the light blue emission band at 469 nm is 367
 within the background noise of detection at the low laser power 368
 density $\Phi_{\text{laser}} = 1.02 \text{ MW cm}^{-2}$. This laser power density was 369

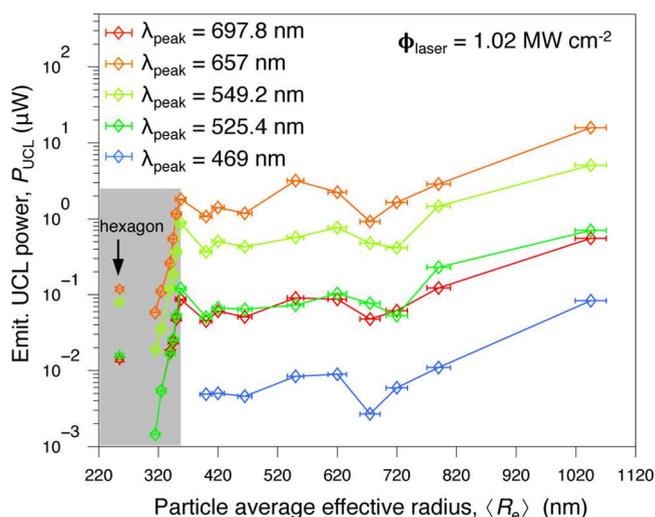


Figure 5. UCL power emitted by a trapped UCP as a function of its average effective radius ($\langle R_e \rangle$). Only the five major peaks are represented (Figure 3c). The gray-shaded area highlights the region where P_{UCL} strongly depends on $\langle R_e \rangle$. Error bars originate from the uncertainty in reading out τ_{fir} .

employed to probe $\langle R_e \rangle$ for all particles investigated in Figure 5. The very steep increase of P_{UCL} below $\langle R_e \rangle = 357.5$ nm (Figure 5a, gray-shaded area) can be explained by the ratio of the particle and laser beam sizes at the trapping position. As long as $\langle R_e \rangle < w_0$, with $w_0 = 451$ nm being the $1/e^2$ radius of the focused trapping laser,²¹ any radial variation in the excitation density has drastic effects on P_{UCL} .²⁶ Furthermore, defect-induced quenching becomes more important for smaller ratios, $\langle R_e \rangle/w_0$. Such quenching is clearly higher for a random shape and polycrystalline UCP than for a monocrystalline and hexagonal shape UCP with well-defined edges, which exhibits therefore a more efficient upconversion process, despite its smaller size^{10,35,36} (Figure 5). As soon as $\langle R_e \rangle \gtrsim w_0$, the UCP sizes begin to exceed the laser beam diameter and then the thermal fluctuations influence less P_{UCL} (Figure 5, plateau at $\langle R_e \rangle > 357.5$ nm).

The results presented in Figure 5 can also allow tailoring the best-suited UCP morphologies for performing energy transfer in fluorescence microscopy from a single UCP to an adjacent chromophore with subwavelength resolution. In particular, the appropriate size of UCP has to be comparable to the size of the emission volume in a two-photon excitation (TPE) process, directly induced by the focused laser beam. In the case of our PFM operating at 1064 nm this emission volume is defined by the full width at half-maximum ($\text{fwhm}_0 = 345$ nm and $\text{fwhm}_z = 866$ nm).³⁷ Matching the UCP size and the TPE volume enables the energy transfer to dominate oversimple TPE processes. To further improve the spatial resolution, tuning the UCL emission by designing core/shell-structured upconversion particles is worth investigating.^{38,36}

CONCLUSIONS

In summary, we characterized in detail, at the single-particle level, the upconversion properties of random and hexagonal shape particles of $\text{NaYF}_4:\text{Yb,Er}$. These nano/microsized UCPs could be easily manipulated and revealed efficient nonlinear multicolor UCL, while being trapped by the NIR laser beam of the PFM. Our results indicate also that the amplitudes of individual visible UCL emission bands depend on the laser

power density as well as the size and shape of the particles. In particular, the hexagonal plates exhibit a much stronger UCL emission than random shape particles of comparable sizes. This might be due to a lower content of structural defects in the highly crystalline structures. Moreover, under high laser power densities UCPs were photostable and revealed no blinking. In addition, the analysis of the Brownian motion of the trapped UCP allowed us to precisely measure its two-dimensional sizes. This was done by exploiting the short time dynamics of thermal fluctuations of the UCP through the simultaneous fitting of the VAF and PSD.

In brief, further research in this domain will broaden the application field of $\text{NaYF}_4:\text{Yb,Er}$ nanoparticles from photoluminescent single-particle probes to such applications as nanolight sources for multiplexed molecule sensing at the nanoscale with subwavelength resolution. In biomedically oriented studies, the novel microscopic methodology proposed herein can provide a deeper insight into mechanical and microrheological properties of intracellular components and interactions. In particular, due to the implementation of PFM, stiffness, viscosity, and exerted forces can be evaluated with nanometer spatial and microsecond temporal resolutions.

ASSOCIATED CONTENT

Supporting Information

Further details on the PFM instrument (Figure S1), morphologies of the UCPs, optical trapping of single UCPs, laser-induced heating in PFM, calibration of the spectrofluorometer (Figures S2 and S3), quantification of the UCL spectrum (Figure S4), and UCL bleaching (Figure S5) can be found in the Supporting Information. This material is available free of charge via the Internet at <http://pubs.acs.org>.

AUTHOR INFORMATION

Corresponding Author

*E-mail: flavio.mor@epfl.ch. Phone: +41 (21) 693 4438. Fax: +41 (21) 693 4470.

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

We thank S. Haacke and H. U. Güdel for useful discussions. S.J. and F.M.M. acknowledge support from the National Centre of Competence in Research (NCCR Nano-Bio, Project 1.4) and the Swiss National Science Foundation (SNSF, project numbers 206021_121396 and 200021_143703). F.M.M. was supported by the National Competence Center in Biomedical Imaging (NCCBI). The research was also cofinanced by the SNSF through the Nano-Tera.ch focused NTF project "NanoUp".

REFERENCES

- (1) Nakayama, Y.; Pauzuskie, P. J.; Radenovic, A.; Onorato, R. M.; Saykally, R.; Liphardt, J.; Yang, P. Tunable nanowire nonlinear optical probe. *Nature* **2007**, *447*, 1098–1102.
- (2) Dutto, F.; Raillon, C.; Schenk, K.; Radenovic, A. Nonlinear optical response in single alkaline niobate nanowires. *Nano Lett.* **2011**, *11*, 2517–2521.
- (3) Reece, P. J.; Paiman, S.; Abdul-Nabi, O.; Gao, Q.; Gal, M.; Tan, H. H.; Jagadish, C. Combined optical trapping and microphotoluminescence of single InP nanowires. *Appl. Phys. Lett.* **2009**, *95*, 1–3.
- (4) Cohen, B. E. Beyond fluorescence. *Nature* **2010**, *467*, 407–408.

- 466 (5) Gorris, H. H.; Wolfbeis, O. S. Photon-upconverting nanoparticles
467 for optical encoding and multiplexing of cells, biomolecules, and
468 microspheres. *Angew. Chem., Int. Ed.* **2013**, *52*, 3584–3600.
- 469 (6) Esipova, T. V.; Ye, X.; Collins, J. E.; Sakadžić, S.; Mandeville, E.
470 T.; Murray, C. B.; Vinogradov, S. A. Dendritic upconverting
471 nanoparticles enable in vivo multiphoton microscopy with low-
472 power continuous wave sources. *Proc. Natl. Acad. Sci. U.S.A.* **2012**, *109*,
473 20826–20831.
- 474 (7) Pantazis, P.; Maloney, J.; Wu, D.; Fraser, S. E. Second harmonic
475 generating (SHG) nanoprobes for in vivo imaging. *Proc. Natl. Acad.*
476 *Sci. U.S.A.* **2010**, *107*, 14535–14540.
- 477 (8) Mader, H. S.; Kele, P.; Saleh, S. M.; Wolfbeis, O. S. Upconverting
478 luminescent nanoparticles for use in bioconjugation and bioimaging.
479 *Curr. Opin. Chem. Biol.* **2010**, *14*, 582–596.
- 480 (9) Soukka, T.; Kuningas, K.; Rantanen, T.; Haasslahti, V.; Lövgren,
481 T. Photochemical characterization of up-converting inorganic
482 lanthanide phosphors as potential labels. *J. Fluoresc.* **2005**, *15*, 513–
483 528.
- 484 (10) Ye, X.; Collins, J. E.; Kang, Y.; Chen, J.; Chen, D. T. N.; Yodh,
485 A. G.; Murray, C. B. Morphologically controlled synthesis of colloidal
486 upconversion nanophosphors and their shape-directed self-assembly.
487 *Proc. Natl. Acad. Sci. U.S.A.* **2010**, *107*, 22430–22435.
- 488 (11) Wang, F.; Liu, X. Recent advances in the chemistry of
489 lanthanide-doped upconversion nanocrystals. *Chem. Soc. Rev.* **2009**, *38*,
490 976–989.
- 491 (12) Wang, F.; Banerjee, D.; Liu, Y.; Chen, X.; Liu, X. Upconversion
492 nanoparticles in biological labeling, imaging, and therapy. *Analyst*
493 **2010**, *135*, 1839–1854.
- 494 (13) Schietinger, S.; de S. Menezes, L.; Lauritzen, B.; Benson, O.
495 Observation of size dependence in multicolor upconversion in single
496 Yb^{3+} , Er^{3+} codoped NaYF_4 nanocrystals. *Nano Lett.* **2009**, *9*, 2477–
497 2481.
- 498 (14) Wu, S.; Han, G.; Milliron, D. J.; Aloni, S.; Altoe, V.; Talapin, D.
499 V.; Cohen, B. E.; Schuck, P. J. Non-blinking and photostable
500 upconverted luminescence from single lanthanide-doped nanocrystals.
501 *Proc. Natl. Acad. Sci. U.S.A.* **2009**, *106*, 10917–10921.
- 502 (15) Schietinger, S.; Aichele, T.; Wang, H.-Q.; Nann, T.; Benson, O.
503 Plasmon-enhanced upconversion in single $\text{NaYF}_4:\text{Yb}^{3+}/\text{Er}^{3+}$ codoped
504 nanocrystals. *Nano Lett.* **2010**, *10*, 134–138.
- 505 (16) Gargas, D. J.; Chan, E. M.; Ostrowski, A. D.; Aloni, S.; Altoe, M.
506 V. P.; Barnard, E. S.; Sanii, B.; Urban, J. J.; Miliron, D. J.; Cohen, B. E.;
507 Schuck, P. J. Engineering bright sub-10-nm upconverting nanocrystals
508 for single-molecule imaging. *Nat. Nanotechnol.* **2014**, *9*, 300–305.
- 509 (17) Haro-González, P.; del Rosal, B.; Maestro, L. M.; Rodríguez, E.
510 M.; Naccache, R.; Capobianco, J. A.; Dholakia, K.; Solé, J. G.; Jaque, D.
511 Optical trapping of $\text{NaYF}_4:\text{Er}^{3+},\text{Yb}^{3+}$ upconverting fluorescent nano-
512 particles. *Nanoscale* **2013**, *5*, 12192–12199.
- 513 (18) Gittes, F.; Schmidt, C. F. Interference model for back-focal-
514 plane displacement detection in optical tweezers. *Opt. Lett.* **1998**, *23*,
515 7–9.
- 516 (19) Franosch, T.; Grimm, M.; Belushkin, M.; Mor, F. M.; Foffi, G.;
517 Forró, L.; Jeney, S. Resonances arising from hydrodynamic memory in
518 Brownian motion. *Nature* **2011**, *478*, 85–88.
- 519 (20) Jeney, S.; Mor, F.; Köszali, R.; Forró, L.; Moy, V. T. Monitoring
520 ligand-receptor interactions by photonic force microscopy. *Nano-*
521 *technology* **2010**, *21*, 1–8.
- 522 (21) Smith, S. P.; Bhalotra, S. R.; Brody, A. L.; Brown, B. L.; Boyda,
523 E. K.; Prentiss, M. Inexpensive optical tweezers for undergraduate
524 laboratories. *Am. J. Phys.* **1999**, *67*, 26–35.
- 525 (22) Peterman, E. J. G.; Gittes, F.; Schmidt, C. F. Laser-induced
526 heating in optical traps. *Biophys. J.* **2003**, *84*, 1308–1316.
- 527 (23) Renero-Lacuna, C.; Martn-Rodríguez, R.; Valiente, R.; Gonzáles,
528 J.; Rodríguez, F.; Krämer, K. W.; Güdel, H. U. Origin of the high
529 upconversion green luminescence efficiency in $\beta\text{-NaYF}_4:2\%\text{Er}^{3+},20\%$
530 Yb^{3+} . *Chem. Mater.* **2011**, *23*, 3442–3448.
- 531 (24) Suyver, J. F.; Aebischer, A.; Garca-Revilla, S.; Gerner, P.; Güdel,
532 H. U. Anomalous power dependence of sensitized upconversion
533 luminescence. *Phys. Rev. B* **2005**, *71*, 125123–125131.
- (25) Wang, F.; Han, Y.; Lim, C.; Lu, Y.; Wang, J.; Xu, J.; Chen, H.;
534 Zhang, C.; Hong, M.; Liu, X. Simultaneous phase and size control of
535 upconversion nanocrystals through lanthanide doping. *Nature* **2010**,
536 463, 1061–1065. 537
- (26) Pollnau, M.; Gamelin, D. R.; Lüthi, S. R.; Güdel, H. U.; Hehlen,
538 M. P. Power dependence of upconversion luminescence in lanthanide
539 and transition-metal-ion systems. *Phys. Rev. E* **2000**, *61*, 3337–3346. 540
- (27) Shan, J.; Uddi, M.; Wei, R.; Yao, N.; Ju, Y. The hidden effects of
541 particle shape and criteria for evaluating the upconversion
542 luminescence of the lanthanide doped nanophosphors. *J. Phys. Chem.*
543 *C* **2010**, *114*, 2452–2461. 544
- (28) Jauffred, L.; Oddershede, L. B. Two-photon quantum dot
545 excitation during optical trapping. *Nano Lett.* **2010**, *10*, 1927–1930. 546
- (29) Patterson, G. H.; Piston, D. W. Photobleaching in two-photon
547 excitation microscopy. *Biophys. J.* **2000**, *78*, 2159–2162. 548
- (30) Friese, M. E. J.; Nieminen, T. A.; Heckenberg, N. R.;
549 Rubinsztein-Dunlop, H. Optical alignment and spinning of laser-
550 trapped microscopic particles. *Nature* **1998**, *394*, 348–350. 551
- (31) Higurashi, E.; Ukita, H.; Tanaka, H.; Ohguchi, O. Optically
552 induced rotation of anisotropic microobjects fabricated by surface
553 micromachining. *Appl. Phys. Lett.* **1994**, *64*, 2209–2210. 554
- (32) Han, Y.; Alsayed, A. M.; Nobili, M.; Zhang, J.; Lubensky, T. C.;
555 Yodh, A. G. Brownian motion of an ellipsoid. *Science* **2006**, *314*, 626–
556 630. 557
- (33) Maragó, O. M.; Jones, P. H.; Bonaccorso, F.; Scardaci, V.;
558 Gucciardì, P. G.; Rozhin, A. G.; Ferrari, A. C. Femtonewton force
559 sensing with optically trapped nanotubes. *Nano Lett.* **2008**, *8*, 3211–
560 3216. 561
- (34) Maragó, O. M.; Bonaccorso, F.; Saija, R.; Privitera, G.;
562 Gucciardì, P. G.; Lati, M. A.; Calogero, G.; Jones, P. H.; Borghese,
563 F.; Denti, P.; Nicolosi, V.; Ferrari, A. C. Brownian motion of graphene.
564 *ACS Nano* **2010**, *4*, 7515–7523. 565
- (35) Bai, X.; Song, H.; Pan, G.; Lei, Y.; Wang, T.; Ren, X.; Lu, S.;
566 Dong, B.; Fan, L. Size-dependent upconversion luminescence in $\text{Er}^{3+}/$
567 Yb^{3+} -codoped nanocrystalline yttria: saturation and thermal effects. *J.*
568 *Phys. Chem. C* **2007**, *111*, 13611–13617. 569
- (36) Mai, H.-X.; Zhang, Y.-W.; Sun, L.-D.; Yan, C.-H. Highly efficient
570 multicolor up-conversion emissions and their mechanisms of
571 monodisperse $\text{NaYF}_4:\text{Yb},\text{Er}$ core and core/shell-structured nanocryst-
572 als. *J. Phys. Chem. C* **2007**, *111*, 13721–13729. 573
- (37) Zipfel, W. R.; Williams, R. M.; Webb, W. W. Nonlinear magic:
574 multiphoton microscopy in the biosciences. *Nat. Biotechnol.* **2003**, *21*,
575 1369–1377. 576
- (38) Yi, G.-S.; Chow, G.-M. Water-soluble $\text{NaYF}_4:\text{Yb},\text{Er}(\text{Tm})/$
577 $\text{NaYF}_4/\text{polymercore/shell/shell}$ nanoparticles with significant en-
578 hancement of upconversion fluorescence. *Chem. Mater.* **2007**, *19*,
579 341–343. 580